

The Growth Mechanism Of Zno Nanorods And The Effects Of Growth Conditions

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Abstract: (ZnO) nanorods were manufactured using a seed layer in a low-temperature solution processing hydrothermal technique, and the effects of varied synthesis parameters on the development of ZnO nanorods were studied. The impact of growth solution concentration on the average diameter of ZnO nanorods shows that when the precursor concentration increases, the average diameter grows. The influence of the seed layer may be noticed in the fact that the rod diameters dropped as the seed layer thickness rose. The influence of growth, while the shows that only the length of the nanorods has changed considerably after a long period of growth, while the diameter value has remained constant. The influence of growth temperature shows that the rate of chemical reactions increases as the temperature of the solution rises. The temperature of a solution affects its pH, which swings over time as it grows. Increased pH produces an increase in the concentration of OH in the solution, which increases the growth rate to increase. The enhanced nucleation causes the nanorods to expand in diameter.

Introduction: The first publication [1] was on a hydrothermal method for producing ZnO Nanorods and Nanowires. A hydrothermal technique involving the heat breakdown of methenamine and zinc nitrate was used to produce ZnO nanorods on glass and Si substrates in a controlled manner. To begin the growth process, a thin covering of ZnO nanoparticles was produced on the substrate [2]. The OH⁻ ion was prepared using hexamethylenetetramine (HMTA), whereas the Zn⁺²cation was formed in solution using zinc nitrate hexahydrate. Changing the HMTA and zinc salt solution concentration to make hexagonally shaped zinc oxide (ZnO) nanorods could be a viable growth mechanism. Small molecule nucleation and hexagonal nanorod growth/fabrication are the two processes in the production of ZnO nanorods. In aqueous solution (Zn(NO₃)₂.6H₂O), low-solubility compounds precipitate from zinc salt. This study demonstrates how hexagonally shaped ZnO nanorods can be made using zinc nitrate hexahydrate and HMTA (C₆H₁₂N₄). HMT generates OH⁻(hydroxyl ions), which are commonly used to make ZnO nanostructures. When the HMTA reacts with water, it produces ammonia which reacts with water molecules in the solution to form ammonium (NH4+) ions. By combining with four molecules of ammonia(NH₃) in solution, Zn²⁺ ions can form tetrahedral zinc Zn(NH₃)₄²⁺. In our scenario, four ammonia

hydroxyl ions combine with Zn^{+2} to produce the building blocks of the zinc ammonium complex $Zn(NH3)_4^{2+}$, and as the temperature of the refluxing solution rises, four coordinated complex ions, ammonia, and a water molecule change into ZnO. When HMTA is thermally degraded, hydroxyl ions are released, which combine with Zn^{2+} ions to produce ZnO [3]. The following are the reactions that lead to the formation of ZnO:

$(CH_2)_6N_4 + 6H_2O \rightarrow 6HCHO + 4NH_3$	1
$\mathrm{NH}_3 + \mathrm{H}_2\mathrm{O} \rightarrow \mathrm{NH}^{+4} + \mathrm{OH}^{-1}$	2
$\operatorname{Zn}^{+2} + 4\operatorname{NH}_3 \rightarrow \operatorname{Zn}(\operatorname{NH}_3)_4^+$	3
$H_2O \rightarrow H^+ + OH^-$	4
$20\mathrm{H}^{-} + \mathrm{Zn}(\mathrm{NH}_3)_4^+ \rightarrow \mathrm{ZnO} + 4\mathrm{NH}_3 + \mathrm{H}_2\mathrm{O}$	5

Water and heat energy dissociated colloidal $Zn(OH)_2$ precipitate into Zn^{+2} and OH^- ions in solution (i.e., while refluxing). When the quantities of Zn+2 and OH^- ions climb over a certain value, the process of ZnO nuclei precipitation is expected to commence. This is the start of the ZnO nucleation process. Solution approach is a pH-sensitive technology that may customize material characteristics by altering OH- ion concentration and reflux temperature. A thorough experiment is required to fully comprehend the impact of these variables. The rise in system thermal energy may be linked to variations in Zn^{+2} and OH^- ions, as only the HMTA concentration was used as a control parameter in this work to modify ZnO's structure and morphology. The dissociation of the zinc complex $(Zn(OH)_2)$ is aided by raising the temperature of the refluxing bath, resulting in zinc ions that are available for free controlled releases of Zn^{+2} and hydroxide ions (OH⁻).Normally, the following happens immediately:

$\operatorname{Zn}(\operatorname{OH})_2 \to \operatorname{Zn}^{+2} + 2\operatorname{OH}^{-}$	<i>e</i>
$Zn^{+2}Z + 2OH^- \rightarrow ZnO + H_2O$	7

The growth of $Zn(NH_3)^{4+}$ to ZnO is kinetically controlled. ZnO aggregation competes with nucleation for particle size and shape. HMTA serves as a template/cap for hexagonal zinc oxide molecule nanorod development in this study. The freshly formed $Zn(OH)_2$ and $Zn(NH_3)^{4+}$ nuclei are thought to combine and form a spherical core after absorbing enough heat energy from the refluxing pot to allow active ZnO molecules to attach. The active ZnO nuclei that result must also be used as building blocks in the finished product. On the ZnO core cells, primary stems are active molecules that form hexagonal forms [4–7]. The formation of ZnO single-crystalline rods is thought to involve multiple layered structures, as shown in figure 1.



Fig.1:The crystal structure of the obelisk-shaped ZnO nanorods is shown in this illustration[8].

Many factors influence the size and orientation of ZnO nanorods, including precursor concentration, seed layer, growth time, growth temperature, and pH solution.

Influence of Growth Solution Concentration

As the concentration of Zn precursors in a solution rises, so does its chemical potential. The number of nucleation centers on the planet is being increased. The substrate surface balances out the increased potential, resulting in a higher nanorod density per unit area. The density of nanorods, on the other hand, tends to stabilize as concentration rises (saturation).To begin with, when the reagent concentration is extremely high, a large number of tiny nuclei are produced. These nuclei do not expand beyond the size required for further development and disintegrate over time.

Second, after nanorods have developed into their first generation, the continuation of nucleation is less advantageous in terms of energy, as a result, the presence of growth particles in solution is much more likely to contribute to the development of an existing nanorod than a new nucleus. Therefore, the quantity of first-generation nanorods has a significant impact on the saturation density of nanostructures. With increasing reagent concentrations, the fraction of surface covered by nanostructures increases [9]. It's worth noting that the number of nanostructures found in a sample per unit surface area is consistent and approaching saturation. The density, length, and diameter of ZnO nano-microrods are affected by the concentration used during synthesis; at higher concentrations, the ZnO rods become micro-sized and c-axis aligned, resulting in a micro-sized diameter. Additionally, at concentrations greater than 400 mM, the microrod-sized ZnO transforms into a polycrystalline thin film. Wire-like NRs with a diameter of 100 nm and a length of 1.2 m are formed at low aqueous solution concentrations (10 to 25 mM) [10].



Fig2: shows FESEM images showing different concentrations of ZnO nanorods precursor concentration (M) a. 0.05M, b. 0.1M, c. 0.15M.of zinc nitrate hexahydrate and hexamethylenetetramine [11].

Seed Layer's Influence

Researchers used electrolytic deposition, magnetron sputtering, and a zinc acetate colloidal ethanol solution to create seed layers of ZnO. The seed layers, on the other hand, help to produce homogeneous and nano structured coatings with visible vertically arranged nanostructures that are dense. As the film thickness increases, large-diameter spherules and plate-like oriented laterally ZnO nanostructures appear. These formations were prominent in their films, and they were ascribed to lateral grain expansion caused by decreased strain as the film thickness increased. The roughness of the ZnO buffer layer provided energetically acceptable nucleation seed sites for atomic adsorption [12], implying that surface roughness influences favored nucleation sites. Despite the fact that seed sizes stayed nearly constant as coating thickness rose (constant and RMS), rod diameters dropped as seed layer thickness grows, seed/rod number density ratios and rod/seed diameter both drop. When the parent seed is curved, it forms nuclei with different c-axis directions, which influences the direction of the developing rods. The narrow rods connect together to produce larger rods when the parent seeds are compressed [14].



Fig.3:FE-SEM images of nanorods grown on glass surfaces with seed layers of different thicknesses, from the top, a) 40 nm; b) 65nm; c) 100 nm; d) 220 nm; e) 340 nm; f) 650 nm [15].

The Effect of Growth Time

In all of the nanostructures were used in all of the studies below, nanostructures were created on there are two sorts of substrates: pre-cleaned glass and ethanol solution-coated glass. Three samples were taken to investigate the growth time effect. Nanostructures were grown in 30 minutes, 3 hours, and 18 hours. The solution concentration and growth temperature were the same in all three trials, and the parameter values were set in the same way as in the previous iterations.

Regardless of their disordered orientation and dispersion, all crystallites have an aspect ratio of a hexagonal prism. Vertically aligned nanorods of a comparable diameter were detected in the 3h

experiment. This suggests that all nanostructures of varying sizes and orientations were suppressed by competing growth. When the development duration of the nanostructures is raised to 18 hours, pyramidal nanorods with an unfinished tip are found, implying that the Zn ions have been exhausted at the end of the growth phase even after such a lengthy time of development. Only the nanorods' lengths have changed dramatically, but the diameter has stayed constant. This reveals that the seed layer's origin has the greatest influence on the diameter of the nanostructures generated [16].



Fig.4:FE-SEM images of ZnO nanowires generated for a 30 min growth time (a) ,1h(b) ,2h(c), 3 h (d), 5 h (e) ,a nd6h(f), respectively. Scale bar: 1 μm [17].



Fig.5:ZnO nanowires' average length as a function of growth time [17].

Influence of Growth Temperature

As the temperature of the fluid rises, chemical processes accelerate. HMTA decomposes fast, producing the required quantity of OH- ions in a short amount of time. This causes greater nucleation and development, with the nanorods' base diameter being bigger than at low temperatures, with both lateral and axial growth occurring very quickly. Rapid nanostructure development, on the other hand, depletes Zn ion stores, rendering full-shaped rods impractical and resulting in incomplete, needle-shaped, or pyramidal-shaped structures. The nanostructure density is large in this scenario due to the higher surface mobility of the ad atoms [16, 18].



Fig.6: FE-SEM images of ZnO nanowires grown at: (a) 70°C, (b) 80°C , and (c) 90°C, respectively. Scale bar: 300 nm [17].



Fig.7: ZnO nanowire length versus growing temperature [17].

Influence of Solution PH

During the conversion, the solution contains three kinds of Zn-containing particles: soluble Zn, Zn(OH)₂, and ZnO. Furthermore, because Zn(OH)₂ has a low solubility in ZnO [19], the soluble Zn concentration is insignificant at low pH values, such as those found in aqueous solutions, and the dissolution precipitation process is inhibited. Dehydration and intrinsic atom rearrangement dominate the in-situ crystallization (phase transition) process, resulting in large diameter and short length nanorods. The conversion process is quite slow, taking several hours in this case [20].Increasing the solubility of Zn(OH)₂ and thus the soluble Zn concentration raises the pH of the solution until the dissolution and secondary precipitation processes take over at high pH values. The pH of the solution is raised by increasing the solubility of Zn(OH)₂ and, as a result, the concentrations of dissolved Zn until the secondary precipitation due to dissolution process takes over at high pH values. The supply of Zn ions is soon depleted as a result of this rapid development, resulting in the construction of incomplete structures with a needle-like tip. It's worth mentioning that the pH of the working solution isn't always neutral, and OH- ions are always present, therefore both methods of determining the contents of a solution are conceivable. As a result, additional growth factors influence the main growth process and, as a result, the form of the nanostructure that is created.



Fig.8. : FE-SEM images of ZnO nanostructures produced at various pH levels in solution: (a) pH = 5.7, (b)pH = 9, (c) pH = 12 [14].

Conclusion

The findings in this research reveal that the synthesis of ZnO nanorods is influenced by the growth mechanism and the influence of growing conditions.By adjusting parameters such as solution concentration, seed layer, growth duration, growth temperature, and pH solution, we explored ZnO nanorods under morphological control. This parameter has been shown to be most sensitive to the surface-to-volume ratio and the shape of ZnO nanorods. Furthermore, the shape and size of the ZnO nanostructures were shown to be affected by the precursor concentration, growth period, and temperature. By adjusting these conditions, it is possible to control the interior structure.

Reference

[1] M, Andres Vergés, A. Mifsud and C. J. Serna, Chem. Soc.Faraday Trans. 86, 959(1990).

[2] L. Vayssieres, K. Keis and S. E. Lindquist, J. Phys. Chem. B.105, 3350(2001).

[3] L. Schmidt-Mende and J. L. MacManus-Driscoll, Mater.Today.10, 40(2007).

[4] R. Wahab, S. G, Ansari, Y. S. Kim, H. K. Seo and H. S. Shin HS, Appl

Surf Sci 253:7622(2007).

[5] R. Wahab, S. G. Ansari, Y. S. Kim, G. Khang and H. S. Shin, Appl

Surf Sci 254:2037(2008).

[6] R. Wahab, S. G. Ansari, H. K. Seo, Y. S. Kim, E. K. Suh and H. S. Shin, Solid State Sci 11:439(2009).

[7] R. Wahab, Y.S. Kim and H.S. Shin, Mater Trans 8:2092(2009).

[8] Z.Wang, X. F. Qian, J. Yin and Z. K. Zhu, J. Solid State Chem. 177, 2148(200).

[9] L. Vayssieres, K. Keis, S. E. Lindquist, and A. Hagfeldt, "Purpose-built anisotropie metal oxide material: 3D highly oriented microrod array of ZnO," Journal of Physical Chemistry B, vol. 105, no. 17, pp. 3350–3352(2001).

[10] L. Vayssieres, "Growth of arrayed nanorods and nanowires of ZnO from aqueous solutions," Advanced Materials, vol.15, no. 5, pp. 464–466(2003).

[11] A. A. Abuelsamen, Shahrom Mahmud, Azman Seeni, Noor Haida Mohd Kaus and O. F. Farhat, "Effects of precursor concentrations on the optical and morphological properties of ZnO nanorods on glass substrate for UV photodetector," Vol. 111, P. 536-545(2017).

[12] H. C. Hsu, Y. K. Tseng, H. M. Cheng, J. H. Kuo, and W. F. Hsieh, J. Cryst. Growth 26, 520(2004).
[13] Y.J. Lee, T.L. Sounart, D.A. Scrymgeour, J.A. Voigt, J.W.P. Hsu, "Control of ZnO nanorod array alignment synthesized via seeded solution growth," J. Cryst. Growth, 304, 80(2007).
[14] Q. Li, V. Kumar, Y. Li, H. Zhang, T.J.Marks, R.P.H. Chang, "Fabrication of ZnO nanorods and nanotubes in aqueous solutions," Chem. Mater.17, 1001(3005).

[15] Berrin Ikizler and M. Sumer Peker, "Effect of the seed layer thickness on the stability of ZnO nanorod arrays," Thin Solid Films 558, 149–159(2014).

[16] Vjaceslavs Gerbreders, Marina Krasovska, Eriks Sledevskis, Andrejs Gerbreders, Irena Mihailova, Edmunds Tamanis and Andrejs Ogurcov, "Hydrothermal synthesis of ZnO nanostructures with controllable morphology change," Royal Society of Chemistry, Issue 8(2020).

[17] Clotaire Chevalier-Cesar, Martine Capochichi-Gnambodoe and Yamin Leprince-Wang, "Growth mechanism studies of ZnO nanowire arrays via hydrothermal method," Appl Phys A, 115:953–960(2014).
[18] M. Guo, P. Diao, S. Cai, J. Solid State Chem. 178, 1864 (2005).

[19] R. Wahab, S. G. Ansari, Y. S. Kim, M. Song and H. S. Shin, .Appl. Surf. Sci.255, 4891–4896(2009).[20]
J. Wang, P. Ma and L. Xiang, Mater. Lett.141, 118–121(2015).